

Modeling of Load Transfer Behavior of AR-Glass-Rovings in Textile Reinforced Concrete

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Synopsis: The failure mechanisms of textile reinforced concrete (TRC), which is a composite of bundles of long fibers and fine-grained concrete, are complex. Even after fracture of the constituents AR-glass and concrete the bond behavior has an important influence on the effective material behavior of the composite. Therefore, any realistic numerical simulation of the TRC composite behavior requires an accurate model of the bond between the reinforcement and the matrix as well as between the filaments within the roving. This paper summarizes the main characteristics of TRC. Further, an adhesive cross linkage approach is used to model the different bond aspects within the roving. In addition to this analytical approach some numerical simulations are presented.

Keywords: adhesive cross linkage; bond behavior; roving model; textile; textile reinforced concrete

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INTRODUCTION

Intensive efforts are currently underway to use *textile reinforced concrete* (TRC) for the manufacturing of novel structural elements and the strengthening of existing structures. In contrast to concrete reinforced with randomly aligned short fibers, TRC is a composite made of continuous rovings and fine-grained concrete (aggregate size less than 1mm=0.04 in). A roving is a bundle of a large number of filaments (Fig. 1), which acts as tensile reinforcement of the brittle concrete.

The rovings are aligned with the principal directions of the internal forces resulting from the expected load. The filaments (continuous fibers) used in this study are made of AR-glass, which is much less costly than other candidate materials like carbon. Textile fabrics can be manufactured to satisfy special requirements such as volume fraction or alignment of reinforcement (Fig. 1). In addition to increasing the stiffness the rovings enhance the ductility of the composite significantly (Fig. 2). New manufacturing methods and applications of TRC require the development and improvement of computational models.

MATERIAL BEHAVIOR OF TRC

The material behavior of the AR-glass fibers is shown in Fig. 3. The filament exhibits a nearly linear elastic behavior up to fracture. The effective bundle behavior of a roving is less stiff compared with that of a single filament, because of the non-uniform stress distribution within the roving. A more ductile behavior can be observed for the textile structure.

Structures made of textile reinforced concrete may fail in different ways, as experiments conducted at the Collaborative Research Center “Textile Reinforcement for Structural Strengthening and Retrofitting” (SFB 528) in Dresden have shown [1]. In addition to the delamination of the TRC layer from an existing concrete structure, several failure modes of the TRC itself are possible [2]. The most important of these are:

- development of microcracks
- accumulation of microcracks into a macrocrack
- debonding of fiber reinforcement from the surrounding fine-grade concrete
- slip between filaments within the roving and between filaments and matrix
- fiber fracture.

In general, the final type of failure of the composite is a combination of the different individual failure modes. Textile reinforced concrete in tension has two different types of failure, depending on the amount of fibers. In specimens with low fiber volume fractions (less than 0.1%) only one matrix crack occurs, and the composite fails by fiber fracture or pullout. Figure 4a) shows a typical force-displacement curve for this type of failure. In specimens with high fiber volume fractions (more than 1%) multiple matrix cracks develop prior to final rupture. After the final crack pattern has developed, the remaining stiffness of the tension specimen depends only on the stiffness of the fibers augmented by tension stiffening. A typical force-displacement curve of this type is shown in Fig. 4b). For both types of TRC the bond behaviors between roving and matrix and within the roving are decisive in determining the overall structural behaviour.

Figure 5 shows two cross sections of rovings without any textile binding embedded in fine-grained concrete. Here we want to point out the partial impregnation of the rovings and the different shapes of the roving cross sections. For a small tex value (e.g. 155 tex) the roving is nearly circular. With increasing tex value the cross sectional shape becomes more flat oval or rectangular due to manufacturing and application issues. Their bond properties are influenced by several parameters, such as fiber size, concrete mix design and age.

MOTIVATION FOR THE ADHESIVE CROSS LINKAGE APPROACH

The bond behavior has been investigated using an environmental scanning electron microscope (ESEM). The micrographs obtained by Hempel et al. [3] (Fig. 6) show the discontinuous distribution of hydration products between the glass filaments which are responsible for the roving's bond properties. These discrete regions with hydration products which provide bond between adjacent filaments shall be referred to as adhesive cross linkages [5]. The distribution of these linkages over the cross section of a roving has a decisive influence on the bond behavior of the roving. Since the roving acts like a filter for the hydrated cement particles, it can be assumed, that the number of adhesive cross linkages decreases towards the center of the roving (Fig. 7b). The strengths of these cross linkages themselves are subject to statistical scatter.

To model the adhesive cross linkages, their actual random distribution (Fig. 8a) is

idealized by lumping them together into a continuous “plug”, leaving a “free length” without any bond (Fig. 8b). Individual filaments are thus restrained according to their relative positions within the roving.

If a roving embedded within the matrix is subjected to tension, the different free lengths $l_w(r)$ of the various filaments cause higher stresses in the filaments on the periphery of the roving (sleeve filaments) (Fig. 9a). The filament stresses σ_{fil} can be determined with Eqs. 1-2, where E = Young’s modulus of the filament and w = half crack width.

$$\sigma_{fil}(r) = E_{fil} \varepsilon_{fil} \quad (1)$$

$$\sigma_{fil}(r) = E_{fil} \frac{w}{l_w(r)} \quad (2)$$

If the filaments fail in tension (Fig. 9b), the question arises whether the roving-matrix system fails in a brittle manner or a further load increase is possible in spite of the reduced cross sectional area.

To answer this question, different variations of the free filament lengths across the roving section were investigated. These free lengths increase from the roving perimeter towards the center in various ways. This increase may be represented by a parabolic, a linear or an exponential function (Fig. 10). The total tensile force in the roving is then,

$$F(w) = \int \sigma(r) dA = \int_{r=0}^R E \frac{w}{l_w(r)} dA \quad (3)$$

In the special case of a roving with circular cross section and quadratic distribution of the free filament lengths along the radius, Eq. 3 leads with

$$dA = 2\pi r dr \quad (4)$$

to,

$$F(w) = 2\pi E w \int_{r=0}^R \frac{r}{l_w(r)} dr \quad (5)$$

$$F(w) = 2\pi E w \int_{r=0}^R \frac{r}{l_{max} - \frac{\Delta l_w}{R_0^2} r^2} dr \quad (6)$$

$$F(w) = w \frac{\pi R_0^2 E}{\Delta l_w} \ln \left(\frac{l_{max}}{l_{max} - \Delta l_w \left(\frac{R}{R_0} \right)^2} \right) \quad (7)$$

In general, the force F in the roving depends on a function $g(r)$, which reflects the

assumed profile of the free lengths $l_w(r)$ over the relative position r within the roving and the cross sectional shape of the roving,

$$F = c_i \int_0^{r(w)} \frac{dr}{g(r)} \quad (8)$$

Introducing the failure strain ε_F of a filament it is possible to establish a relationship between crack width w and the partially effective roving with residual radius R ,

$$w(R) = \left(l_{\max} - \Delta l_w \left(\frac{R}{R_0} \right)^2 \right) \varepsilon_F = l_R(R) \varepsilon_F \quad (9)$$

The roving force F is then a function of this residual radius R ,

$$F(R) = \pi E \frac{R_0^2}{\Delta l_w} \ln \left(\frac{l_{\max}}{l_R(R)} \right) l_R(R) \varepsilon_F \quad (10)$$

The question of when a roving-matrix system can experience further load increase after failure of the outermost (peripheral) filaments is answered in Fig. 11 for the special case of a circular roving with quadratic distribution of the free lengths. As long as $l_{\max}/l_w < e$, no load increase occurs after failure of the peripheral filaments. In the extreme case of $l_{\max}/l_w = 1$, the maximum failure load F_0 of the fiber-matrix system is reached, i.e. all filaments fail when the crack width reaches the critical value w_R . In the special case of $l_{\max}/l_w = e$, a transition from brittle to quasi-ductile behavior takes place. After failure of the peripheral filaments at the crack width w_R , the crack can open further until all remaining filaments have failed. In the case of $l_{\max}/l_w > e$, a load increase takes place after failure of the peripheral filaments at load F_R .

However, the maximum force F_{\max} is still less than the maximum failure load F_0 (Fig. 11). To determine the effective radius R for this maximum force F_{\max} , Eq. 10 can be solved for R . The optimum value for radius R is found as follows:

$$0 = \frac{\partial}{\partial R} F(R) \quad (11)$$

$$R_{\text{opt}} = \pm \frac{R_0^2}{e \Delta l_w} \sqrt{\left(\frac{\Delta l_w l_{\max}}{R_0^2} (e-1) e \right)} \quad (12)$$

with

$$\Delta l_w = l_{\max} - l_R \quad (13)$$

Thereby the ratio R_{opt}/R can be used to describe the desired quasi-ductility of the engineered roving-matrix-system.

The roving force F relative to the force at first cracking F_R is shown in Fig. 12a as function of the relative crack opening w/w_0 . The more non-uniform load distribution within the roving at increasing l_{\max}/l_w ratio reduces the force necessary to break the peripheral filaments and the maximum roving force, Fig. 12b.

Figure 13 summarizes the l_{\max}/l_R ratios necessary to cause quasi-ductile behavior for circular roving sections and rectangular ones with sides a and b (Fig. 5b). It is noteworthy that for an exponential distribution of the free lengths no load increase is possible after failure of the peripheral filaments.

In addition to the analytical approach some numerical simulations were conducted to verify the chosen assumption. The used numerical models differ in the bond description for the roving-matrix system. Beside a continuous bond model with smeared (effective) interface parameter a discontinuous bond model with randomly distributed adhesive cross linkages will be presented.

CONTINUOUS DESCRIPTION OF BOND

The continuous interface model considers the discrete adhesive cross linkages in a smeared sense by means of an interface layer with effective material parameters (Fig. 14). This model has been extensively studied during the past decades, see e.g. Soh et al.[8]

Here the focus lies on the analytical determined multi-linear bond laws between the modeled layers of the roving. Figure 16 shows a roving of circular section that was modeled with four layers of filaments. Point b is situated at the midpoint between two cracks, and point a at the macrocrack, where the roving is loaded, i.e. where it bridges the crack.

The relationship between load and crack opening has three distinct regions. The first region corresponds to the undamaged state. Subsequent loading causes damage and failure of the interface between the matrix and the outermost (peripheral) layer of filaments.

For the example configuration a linear decrease of bond strength across the roving section was assumed. This bond strength was modeled with a multi-linear bond-slip relationship by Richter [4] and Zastrau et al [6] (Fig. 14a, 15), based on pull-out experiments [7] (Fig. 14b).

After complete separation of the interface 4, the roving is loaded up to failure. In this example, the interfaces 1 to 3 do not fail.

DISCONTINUOUS DESCRIPTION OF BOND

Aside from the continuous bond representation also discontinuous bond models were investigated. Again, the roving was idealised by a number of layers (Fig. 17a). The adhesive cross linkages were represented by elastic bond elements with a specific failure threshold. Assuming the closest possible packing of filaments within a roving of circular

section, a roving of about 155 *tex* (397 filaments) will contain 12 discrete layers.

The computed roving stresses at the crack center depend on the distribution of the adhesive cross linkages. The relative share of linkages in each interface is depicted in Fig. 18. For example, 90% of the interface between the matrix and outermost layer of filaments contain adhesive cross linkages, whereas only 20% of the innermost interface is covered with such linkages.

In Fig. 18, the stress distributions in a roving-matrix system are shown for several realizations of random distributions of the adhesive cross linkages. Since the stresses are inversely proportional to the free length parameter,

$$\sigma(r) \sim \frac{1}{l_w} \quad (15)$$

it is possible to compute the free lengths within the roving for each such random realization. For a given profile free length across the roving the free lengths l_w vary approximately parabolically, Fig. 19a. The resulting statistical distribution of the calculated maximum free length l_{max} for $r/R_o = 0$ is illustrated in Fig. 19b.

SUMMARY

The combination of a number of individual failure mechanisms complicates the material behavior of textile reinforced concrete. The bond between the fibers and the matrix has a considerable influence on this material behavior and is very difficult to model, because of the non-uniform stress distribution over a roving section. This paper presents a model to simulate the effect of the discrete hydration products found between individual filaments (adhesive cross linkages). This model is capable of distinguishing between brittle and quasi-ductile roving failure, depending on the extent of hydration product impregnation within a roving. The validity of the various assumptions for this model was confirmed with finite element analyses using either continuous or discontinuous bond models.

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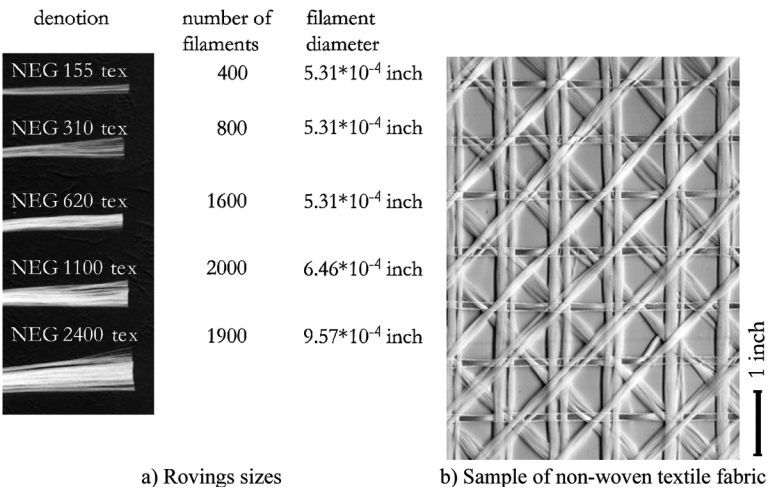


Fig. 1—Textile reinforcement.^[1]

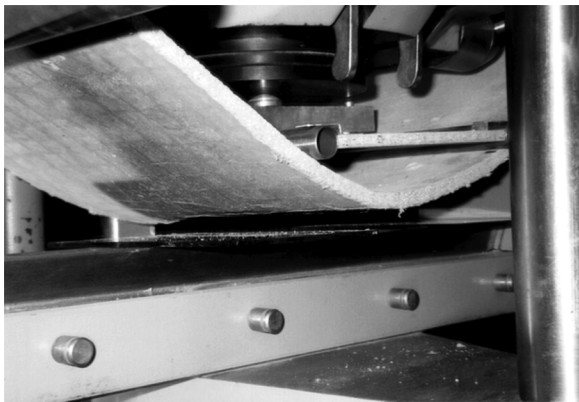


Fig. 2—Bending of a textile reinforced plate.^[1]

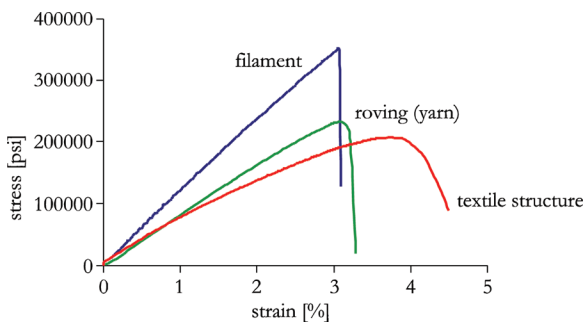


Fig. 3—Material behavior of glass fibers.

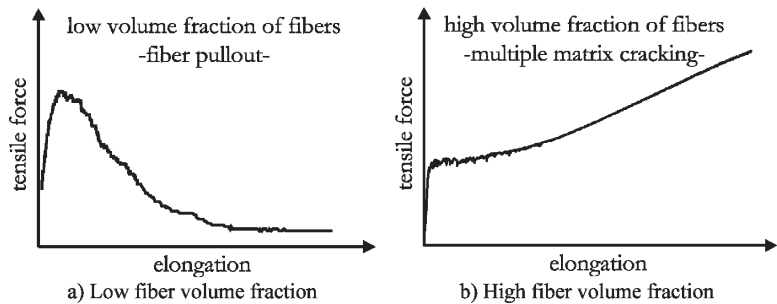


Fig. 4—Force-displacement curves of tensile specimens.

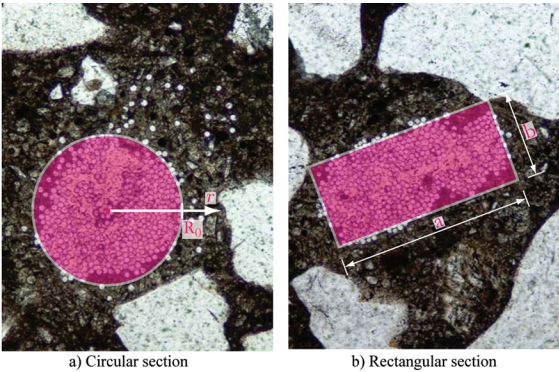


Fig. 5—Sample cross sections of rovings embedded within a cement matrix.

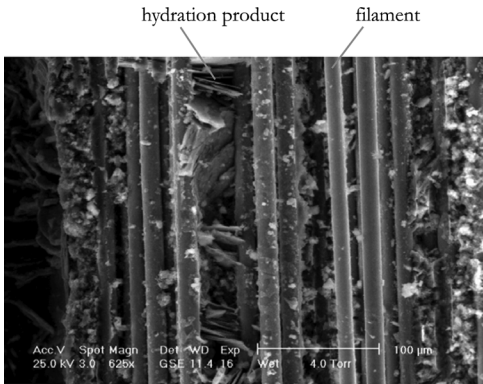


Fig. 6—ESEM Micrograph of roving, longitudinal section.^[3]

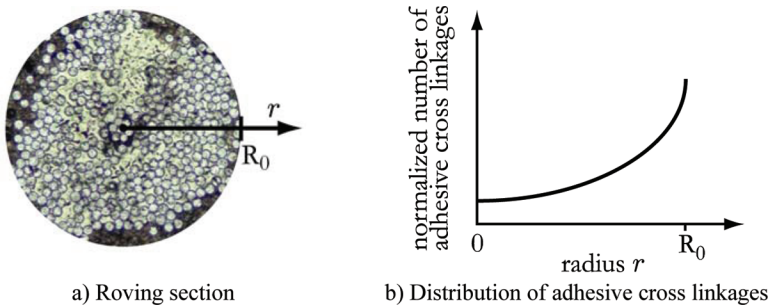


Fig. 7—Distribution of adhesive cross linkages in a circular roving.

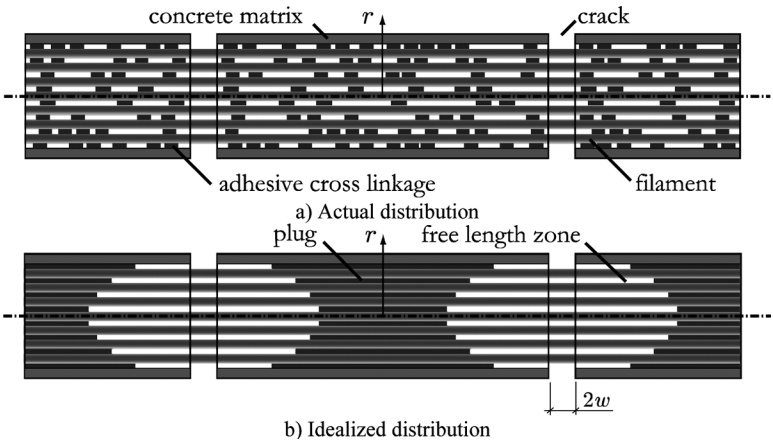


Fig. 8—Plug model.

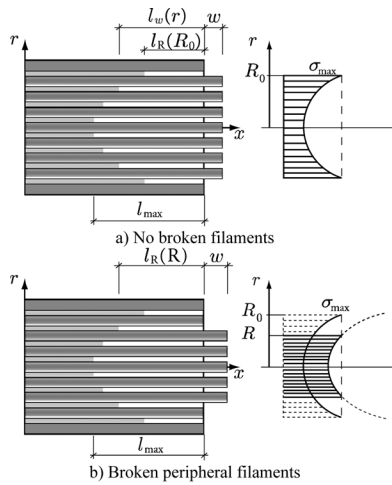


Fig. 9—Stress distributions in roving.

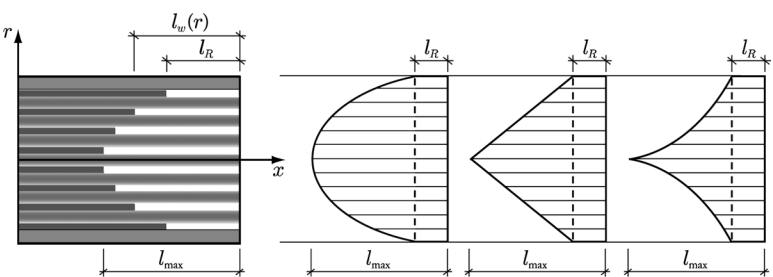


Fig. 10—Possible profiles of free lengths.

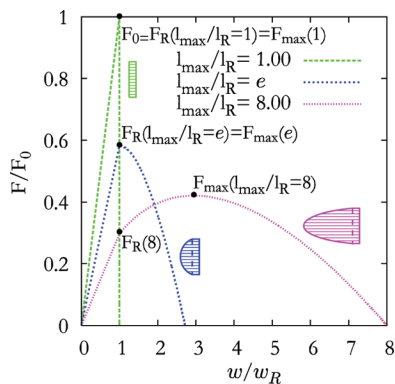


Fig. 11—Roving force versus crack width (circular section, parabolic free length profile).

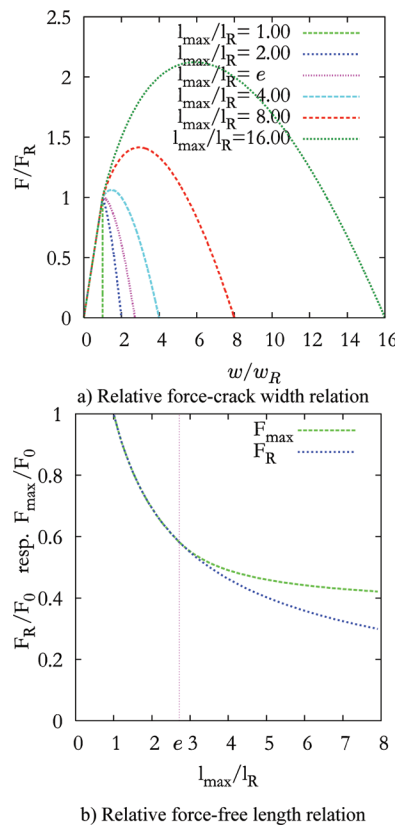


Fig. 12—Related force dependency.






				
	$g(r)$	$\frac{l_{max}}{l_R}$	$g(r)$	$\frac{l_{max}}{l_R}$
	$\frac{a}{r} - b$	≈ 4.93	$a - br$	2.71828
	$\frac{a}{r} - br$	2.71828	$a - br^2$	≈ 1.72
	$\frac{1}{r}e^{(a-r)}$	—	$e^{(a-r)}$	—

Fig. 13— l_{max}/l_R ratios necessary to cause quasi-ductile behavior.

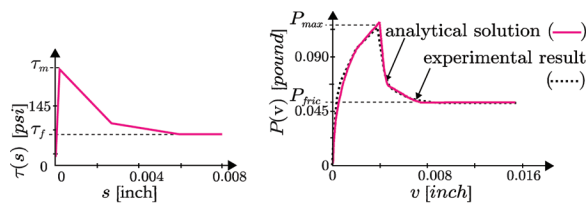


Fig. 14—Shear stress-slip relation and filament pull-out test result.^[4,6,7]

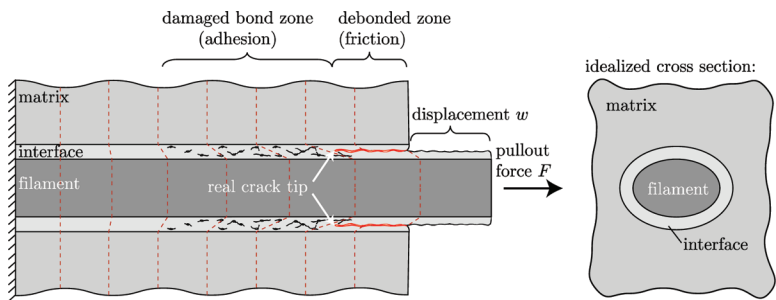


Fig. 15—Bond simulation by means of continuous interface model.

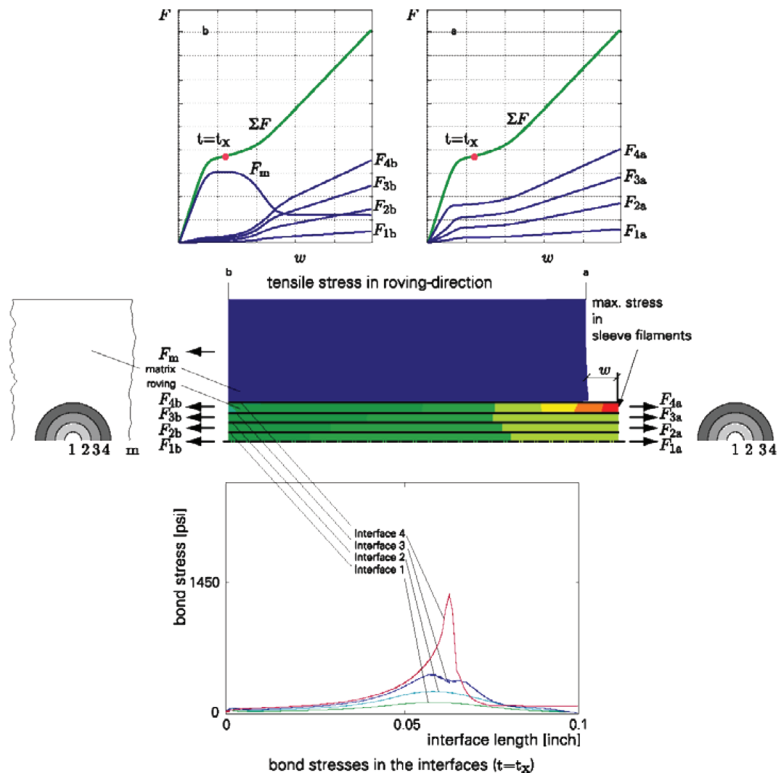


Fig. 16—Continuous bond model.

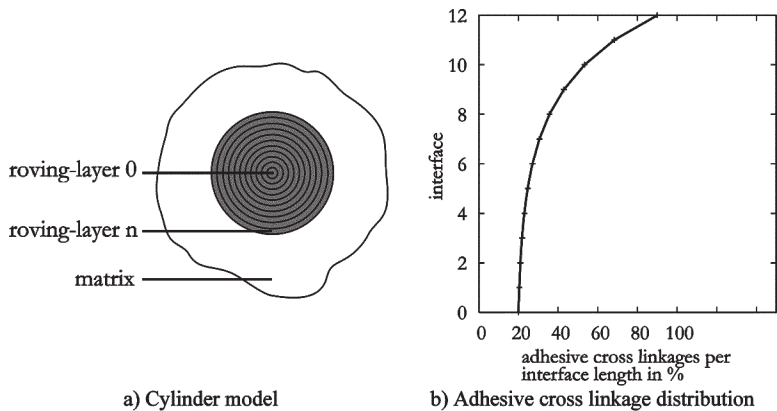


Fig. 17—Cylinder model with discontinuous bond.

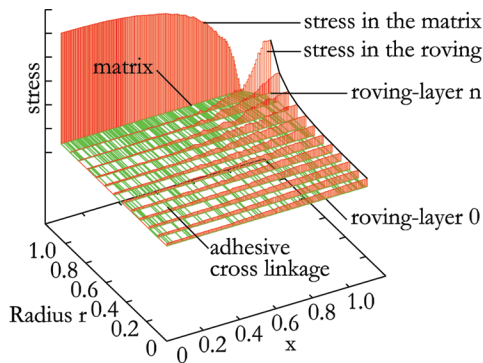


Fig. 18—Filament stresses in model with discontinuous bond.

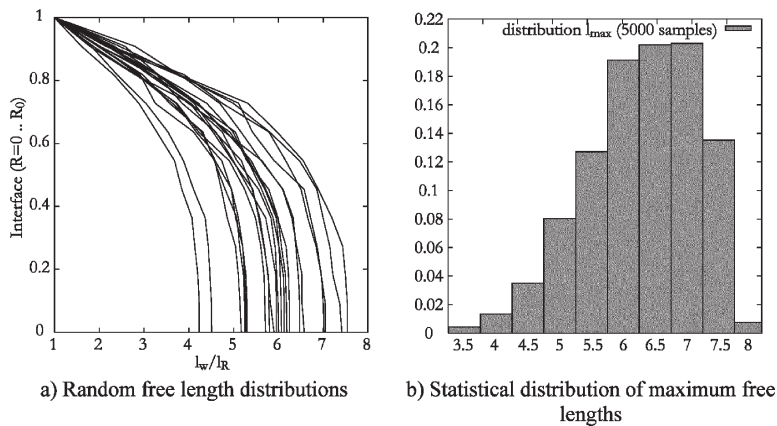


Fig. 19—Statistical evaluation.

